X-ray spectroscopy reveals high symmetry and electronic shell structure of transition-metal-doped silicon clusters

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Size-selected cationic transition-metal-doped silicon clusters have been studied with x-ray absorption spectroscopy at the transition-metal L_{2,3} edges to investigate the local electronic structure of the dopant atoms. For VSi_{16}, the x-ray absorption spectrum is dominated by sharp transitions which directly reveal the formation of a highly symmetric silicon cage around the vanadium atom. In spite of their different number of valence electrons, a nearly identical local electronic structure is found for the dopant atoms in TiSi_{16} VSi_{16}, and CrSi_{16}. This indicates strongly interlinked electronic and geometric properties: while the transition-metal atom imposes a geometric rearrangement on the silicon cluster, the interaction with the highly symmetric silicon cage determines the local electronic structure of the transition-metal dopant.

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I. INTRODUCTION

Doped silicon clusters are perfect examples of tailoring nanoparticle properties by choice of size and composition. Since silicon prefers sp^3 bonding or even higher coordinated bonds reminiscent of its high pressure metallic phases [1], pure silicon clusters tend to form complex and rather compact structures [2–5], but never form cages as carbon does. Incorporation of a single impurity atom, however, rearranges the geometric and electronic structure of small silicon clusters completely. In particular, this can lead to a stabilization of the otherwise unfavorable cage structures [6–9]. Prominent examples are ScSi_{16}, TiSi_{16}, and VSi_{16}, which are highly stable and therefore largely abundant in mass spectra [10–14] because of simultaneous electronic and geometric shell closure [8,9]. Their electronic stability is reflected in a large highest occupied molecular orbital–lowest unoccupied molecular orbital gap of about 2 eV, which has been predicted theoretically [6–9] and was confirmed by photoelectron spectroscopy for the case of TiSi_{16} [14]. The formation of silicon cages can be inferred experimentally from reactivity [14–18], photodissociation [19], and physisorption [20] studies. Indirect evidence is supplied by photoelectron spectroscopy [12,14,16,17,21–24].

Despite a wealth of theoretical and experimental work, the electronic structure and nature of bonding in these silicon cages is still controversial [6–9,12,14,17,24]. Two competing models are commonly used, rationalizing the electronic structure obtained from ab initio calculations [7,8] either on the basis of the empirical 18 (in this case rather 20) electron rule [8,25,26] known for transition-metal compounds in chemistry or on electronic shell closure of delocalized electrons in a spherical potential model combined with an approximate selection rule for the angular momentum [7,9,27,28]. In a simplified description, the first model explains stability on the grounds of electronic shell closure at the transition-metal atom, while the second model considers electronic shell closure in the spherical potential of the entire cluster cage.

II. EXPERIMENTAL SETUP

We have used x-ray absorption spectroscopy on size-selected free clusters [29] as a local and element-specific probe to clarify the local electronic structure of the dopant atom that leads to the exceptional stability of doped silicon cages. Transition-metal-doped silicon cluster cations are produced in a magnetron sputtering source by co-sputtering of two targets. After size selection in a quadrupole mass selector, clusters are stored in a liquid nitrogen cooled linear Paul trap. Along the trap axis, electronic transitions are excited by a collinear beam of tunable soft x-ray radiation. Ions are subsequently extracted from the trap and are analyzed in a time-of-flight mass spectrometer. X-ray absorption spectra are obtained in ion yield mode by monitoring the yield of cluster fragment ions as a function of incident photon energy.

III. RESULTS

Vanadium L_{2,3} x-ray absorption spectra of free, size-selected VSi_{n}^{+} clusters with n=14−18 are presented in Fig. 1. In this size range, VSi_{16} is a prominent line in mass spectra [12,14]. Compared to other clusters, the x-ray absorption spectrum of VSi_{16} exhibits the best resolved structure with sharp lines, which directly indicate a high degeneracy of electronic levels and a high symmetry of this cluster. Adding or removing silicon cage atoms lowers the symmetry and broadens electronic transitions as in VSi_{17} or VSi_{15}. The origin of the distinct double peak structure observed for VSi_{16} in resonant excitation at the L_{3}(2p_{3/2}→3d,4s) and L_{2}(2p_{1/2}→3d,4s) edges will be discussed below.

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When keeping the number of silicon atoms constant but changing the dopant atom from vanadium to titanium or chromium, the number of valence electrons in the cluster is changed by \( \pm 1 \). The resulting x-ray absorption spectra of \( \text{TiSi}_{16}^+ \), \( \text{VSi}_{16}^+ \), and \( \text{CrSi}_{16}^+ \) are shown in the lower panel of Fig. 2. For ease of comparison, the spectra are aligned at the first peak position of the transition-metal \( L_3 \) excitation and are given on a relative photon energy scale. Surprisingly, titanium, vanadium, and chromium exhibit a very similar fine structure in their x-ray absorption spectra when embedded in \( \text{Si}_{16} \) cages; the relative excitation energies are practically identical for all three dopant atoms. Not only the intense lines at 0 and 2.1 eV, but also the less intense features at 1.1 and 5.0 eV are reproduced in all spectra. This indicates a nearly identical local electronic density of states at the transition-metal atoms and a very similar geometry of their environment, as x-ray absorption spectroscopy is very sensitive to both [30]. For comparison, the spectra of the bare transition-metal ions are displayed in the upper panel of Fig. 2. As expected, the multiplet structure is characteristic for each element and reflects the differences in electronic structure, namely, the different 3d orbital occupancies. Apparently, these differences disappear when the atoms are located in \( \text{Si}_{16} \) cages. Here, titanium and chromium adopt the same local electronic structure as vanadium in \( \text{VSi}_{16}^+ \) in spite of their differing number of valence electrons. Obviously, the interaction with the highly symmetric silicon cage determines the electronic structure of the dopant.

X-ray absorption spectroscopy was also performed at the silicon \( L_{2,3} \) edges. As representative examples, the spectra of \( \text{Si}_{16}^+ \) and \( \text{VSi}_{16}^+ \) are compared in Fig. 3. While pure silicon clusters in this size range generally show broad spectra similar to the one of \( \text{Si}_{16}^+ \), the spectrum of \( \text{VSi}_{16}^+ \) shows sharp lines, which indicate a well structured electronic density of states that reflects the high symmetry of the cluster. Although x-ray absorption is not a direct probe of the ground-state electronic structure, and there is no simple one-to-one correspondence of the x-ray absorption spectra to the unoccupied density of states, this again hints at highly degenerate electronic states.

**IV. DISCUSSION**

These findings can be well understood within the spherical potential model [7,9,28,31], which was originally develop-
oped to explain the nature of bonding in pure and endohedral fullerences [27,28,31–33]. A transition-metal-doped Si16 Frank-Kasper polyhedron [6,34] possesses a total of 64 (i.e., 16 × 4) silicon valence electrons; additional valence electrons are supplied by the transition-metal atom hosted at the center of the cage. Because of the nearly spherical shape of the cluster, its valence orbitals can be classified by an angular momentum quantum number ℓ and a radial quantum number n in harmonic oscillator notation. In a simplified view, one can regard bonding within the spherical Si16 shell to be mediated by “σ” orbitals (1s, 1p, 1d, 1f, 1g) without radial nodes [31,35,36], while bonding to the transition-metal dopant is promoted by “π” orbitals (2s, 2p, 2d) with one radial node [31,35,36]. These orbitals hybridize with transition-metal valence orbitals of the same angular momentum quantum number according to the approximate ℓ-selection rule [27,28]. In particular, the transition-metal 4s and 3d orbitals hybridize with silicon cage s and d “π” orbitals to form bonding (2s, 2d) and antibonding (3s, 3d) orbitals. In fact more than these four levels are formed: as the Frank-Kasper Si16 cage [6,34] is not perfectly spherically symmetric but has tetrahedral (Td) symmetry, a d state is represented as (e + t2). Therefore the metal 3d orbitals can interact with all silicon cage orbitals having e or t2 components and do so especially with the energetically close-by 1h (e + t1 + 2t2) states. This leads to the formation of three groups of electronic states with major contributions of metal 3d orbitals [9].

The resulting schematic density of states of VSi16+ [7,9] is shown in Fig. 4: a total of 68 (i.e., 16 × 4 + 5 − 1) valence electrons occupy all states of the σ (n = 1) system up to the 1g level and of the π (n = 2) system up to the 2d level, which forms the highest occupied molecular orbital of the cluster. The lowest unoccupied molecular orbital has 1h character [9] and there is a large 2d − 1h gap [6–9]. As discussed above, the metal atom 3d orbitals contribute to the completely filled cage 2d orbital as well as to two narrow groups of unoccupied levels above the gap, which are about 2 eV apart. Since the vanadium 2p core hole is created locally at the dopant site in the x-ray absorption process, and dipole transitions are allowed from the vanadium 2p core level into unoccupied local orbitals with d or s angular momentum character, these sharp features of the electron density of states show up in the x-ray absorption spectrum of VSi16+ and can explain the double structure of the L3 and L2 lines for this closed shell species.

The similarity of the titanium and chromium spectra can be understood in the same picture: in the case of “nonmagic” open shell TiSi16+ and CrSi16+ cage clusters with 67 (i.e., 16 × 4 + 4 − 1) and 69 (i.e., 16 × 4 + 6 − 1) valence electrons, respectively, the energetic ordering of the orbitals should remain very similar, as the nearly identical spectra in Fig. 2 demonstrate that the silicon cage retains its highly symmetric structure. In CrSi16+ the additional valence electron is then accommodated in the 1h orbital above the 2d−1h gap. Because of the dominant silicon contribution to this 1h orbital [9], the electron can be expected to reside mainly at the silicon cage. The local unoccupied 4s and 3d electronic states of the dopant, which are probed by chromium L2,3 x-ray absorption spectroscopy, are therefore hardly affected by this additional valence electron, for which reason the CrSi16+ and VSi16+ x-ray absorption spectra are nearly identical in their main features. The slight broadening observed for CrSi16+ is probably due to Jahn-Teller deformation because of the open electronic shell situation [37].

Similarly, in the case of TiSi16+ with its 67 valence electrons, the electron missing to the closed electronic shell leads to a 2d0 configuration of the cluster. Since the cluster 2d orbital has most of its weight at the Si16 cage [9], only a fraction of the hole is located at the titanium dopant. Again, the small change in the local 3d occupancy of the transition-metal dopant has a negligible influence on the x-ray absorption spectrum. In the three cases of TiSi16+, VSi16+, and CrSi16+ considered here, the interaction with the electronic structure of the highly symmetric silicon cage results in a nearly identical local electronic structure of the dopant and leads to the very similar x-ray absorption spectra observed experimentally. In this respect, x-ray absorption spectroscopy is complementary to valence band photoelectron studies, because the former gives insight into the local electronic structure while the latter probes global properties.

In principle the explanation given above is very close to the crystal-field splitting model usually applied to explain the x-ray absorption spectra of transition-metal ions in compounds [30]. In VSi16+ the vanadium ion occupies a tetrahedral site at the center of the Si16 cage. This cage is tetravalent because of the four electrons missing to the 2d electron shell closure. The vanadium ion contributes its four valence electrons to bonding orbitals leaving no 3d electrons in localized antibonding states. Formally this leads to a local 3d0 con-
figuration at the vanadium site. This empty 3d orbital is split into \( \epsilon \) and \( t_2 \) states because of the tetrahedral symmetry of its environment \([38]\), which causes the double peak structure of the x-ray absorption spectrum \([30]\). Obviously, the same formal 3d\(^0\) configuration is adopted by the chromium and titanium ions as well. In all three cases the electronic shell structure of the cage causes a formal transfer of the 3d valence electrons from the metal atoms to the silicon cage, and the cage symmetry determines the structure of the x-ray absorption spectra.

Finally, the “magic” behavior of the closed electronic shell cluster VSi\(_{16}\) \([12,14]\) can be related to spherical aromaticity \([36]\): including the vanadium atom valence electrons in the electron count, the Si\(_{16}\) cage has 18 electrons in \( \pi (n = 2) \) states and 50 electrons in \( \sigma (n = 1) \) states. Consequently, both \( \pi \) and \( \sigma \) states follow the \( 2(N+1)^2 \) rule of spherical aromaticity \([36]\), with \( N = 4(1g) \) for \( \sigma \) states and \( N = 2(2d) \) for \( \pi \) states. Indeed, based on nucleus independent chemical shifts \([39]\), cage aromaticity, predicted recently for Si\(_{16}^+\) cages, has been proposed to explain the stability of TiSi\(_{16}\) clusters \([24,40]\) and can now be traced back to closed electronic shells.

V. SUMMARY

To summarize, x-ray absorption spectroscopy reveals nearly identical local electronic states of the dopant atoms in TiSi\(_{10}\), VSi\(_{16}\), and CrSi\(_{16}\), showing how closely electronic and geometric properties are interlinked in highly symmetric doped silicon clusters. This demonstrates that x-ray absorption spectroscopy of size-selected gas phase clusters is a very powerful technique which can contribute to the understanding of electronic structure and bonding in isolated nanoparticles. Because of its element-specific nature, this technique is particularly useful for the study of mixed systems and is very promising in view of doped metal clusters, clusters-adsorbate complexes, or molecules in well defined solvation shells.

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